

1 **Oil Palm Lignin-Derived Laser Scribed Graphene in Neutral**
2 **Electrolyte for High-Performance Microsupercapacitor**
3 **Application**

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36 **Abstract**

37 **Lignin is a renewable natural resource that could be derived from oil palm empty fruit**
38 **bunches.** It has generated significant interest as a precursor in synthesizing graphene
39 as anode and cathode material for supercapacitors. In this paper, we report the
40 synthesis of 3D hierarchical Laser Scribed Graphene (LSG) on a flexible polyimide
41 substrate from lignin extracted from empty fruit bunches (EFB) of oil palm for micro-
42 supercapacitor applications. The intensity and speed of the laser have been tuned to
43 yield densely compacted oil palm lignin LSG at a laser power of 70 % and a speed of
44 30 % (OPL-LSG 7030). **OPL-LSG 7030 possessed lower equivalent series resistance**
45 **of 60.1 Ω and a larger crystalline size of ~31 nm than the rest of the tested samples.**
46 It exhibited exceptional areal capacitance of 30.77 mFcm⁻² at a current density of 0.08
47 mAcm⁻², an energy density of 0.00176 mWhcm⁻² and a power density of 0.25 mWcm⁻²
48 when using a unique neutral PAAS/K₂SO₄ gel electrolyte. It achieved excellent
49 capacitance retention of 88.4 % after 5000 charge/discharge cycles and remarkable
50 mechanical stability of 95 % after 400 bending cycles. Furthermore, electrochemical
51 studies revealed the redox properties of readily available quinone/ hydroquinone in the
52 oil palm lignin, which could be inherited in graphene electrodes through a feasible and
53 affordable approach for flexible green energy storage applications.

54

55 **Keywords: Oil palm lignin, Laser scribed graphene, Microsupercapacitors, Flexible**
56 **devices**

57

58 1.0 Introduction

59 Practical energy storage components for flexible devices are necessary for almost all
60 consumer applications in the 21st century. Supercapacitors are electrical
61 storage devices that utilize either electrolyte adsorption or rapid surface redox
62 reactions for energy storage in flexible electronics [1]. Supercapacitors bridge the gap
63 between an electrolytic capacitor and a battery by combining their extraordinary
64 energy, power density capability and excellent life cycles in an environmentally
65 friendly, sustainable construction [2]. There are three types of supercapacitors: - i.e.,
66 electrochemical double layer capacitor (EDLC), pseudo capacitor, and hybrid
67 supercapacitor combining an EDLC and pseudo capacitor. EDLCs are typically
68 constructed from activated carbon, carbon nanotubes, and graphene, while pseudo
69 capacitors comprise metal oxides and conducting polymers [3], [4]. As charges are
70 stored on the anode and cathode surface of EDLCs, increasing the surface area of the
71 material increases the energy densities.

72 On the other hand, the pseudo capacitor achieves higher energy densities through a
73 faradaic mechanism involving redox reaction in transferring electrons across the
74 electrode [2], [5]. EDLC electrode with limited charge storing capacity is usually paired
75 with redox reactive metal oxides with poor cyclic stability to overcome each of their
76 drawbacks [6], [7]. In recent studies, doping of heteroatoms (O, N, S, B, P) enhanced
77 the electrical, surface, chemical, and mechanical properties of graphene [6], [7]. For
78 instance, Vignesh and his colleagues have successfully synergized nitrogen-doped
79 reduced graphene oxide with cobalt-based metal oxides (ZnCo_2O_4) to achieve a
80 capacitance of 950 Fg^{-1} [8].

81 Eco-friendly green energy storage components free from lithium-ion significantly
82 prevent toxicity and excessive heat to humans, regardless of whether they are used
83 in wearables or disposed of in the future [9]–[11]. Developing energy storage
84 applications using biomass waste has become an area of research for a sustainable
85 carbon source and as a preventive measure against the negative impact of excessive
86 biowaste on the environment [12]–[14].

87 Lignin is one of the most abundant organic resources on earth and a key source of
88 renewable aromas in nature [15]. It contains more than 60% carbon and is mainly
89 generated as a waste by-product in large quantities in pulp mills, oil palm mills, and

90 biorefinery facilities [16], [17]. The amount of lignin produced from these industries
91 exceeds 62 million tons annually in the United States, but only 2% are recycled into
92 other valuable applications [17]. Malaysia, a major palm oil producer, generates 22–
93 23 million tonnes of empty fruit bunches (EFB) [18]. Utilizing lignin as a carbon
94 precursor provides access to the functional groups, i.e., carboxyl, hydroxyl, and
95 carbonyl groups present within its cell. For instance, lignin with functional groups such
96 as benzyl and phenolic groups provides ion storage sites for ion storage and active
97 ion storage reaction sites, which promotes high redox activity in supercapacitors [16].
98 Lignin therefore has a variety of uses when utilized as an energy storage device [16],
99 [19], including lignin sourced from raw wood [20], kraft pulp [21], [22], softwood [17],
100 hardwood [23], [24], crop waste [25], bio-ethanol [26], [27] and cornstalk residue [28].

101 In recent years, researchers have successfully synthesized oil palm lignin-derived
102 graphene from EFB [29]–[31]. Despite severe downsides such as high energy
103 consumption and complex processes, pyrolysis and carbonisation techniques are
104 alternatives to synthesising graphene from oil palm lignin [17]. Furthermore, graphene
105 produced via those processes have extremely small yields and suffers from significant
106 structural damage as a result of high temperatures (>1000 °C) [1]. The recent
107 discovery of cost-effective laser lithography techniques to produce high-quality 3D
108 graphene from other lignin sources has superseded these conventional techniques,
109 which come with significant drawbacks [7], [21], [23], [32], [33]. The highly porous 3D
110 graphene structure obtained through optimized laser parameters provides a larger
111 surface area for better electrochemical activity and electrolyte diffusion [32]. By
112 optimizing the laser scribing of oil palm lignin-based 3D graphene will improve the
113 quality of graphene with minimal process-induced defects. In this regard, adequate
114 modification of the laser power and speed may result in high-quality graphene in terms
115 of porosity and composition without further compromising the structure of the material
116 [17]. In addition, it is critical to make other decisions, such as selecting the electrolyte
117 and the possible window for the electrode material, to maximize the electrode's
118 capability [5]. It has recently been discovered that the application of a dilute acid
119 electrolyte such as H₂SO₄ on an electrode made of lignin results in protonation of the
120 lignin functionalities and forms cleavage between the linkages found on lignin [34],
121 [35], preventing lignin from performing to its maximum capacity.

122 This research demonstrates the robust synthesis of graphene derived from novel
123 biowaste oil palm lignin on a flexible polyimide substrate using a low-cost, precise, and
124 single-step laser scribing technique. Independent variables, including laser power and
125 speed, have been well-optimized to synthesize high-quality graphene without further
126 damaging the flexible substrate. To our knowledge, for the first time, synthesized
127 lignin-derived graphene electrode has been assembled employing a neutral electrolyte
128 (K_2SO_4) combined with sodium polyacrylate gel to maximize the redox potential of
129 hydroquinone found in lignin. An outstanding areal capacitance of 30.77 mFcm^{-2} at
130 0.08 mAcm^{-2} , excellent cyclic stability of 88.4 % capacitance retention rate after 5000
131 charge-discharge cycles, energy density of $0.00176 \text{ mWhcm}^{-2}$ and a power density of
132 0.25 mWcm^{-2} has been achieved by optimized the lignin-derived graphene electrode.
133 Oil palm lignin can be effectively transformed into a functionalized graphene electrode
134 for flexible micro-supercapacitor applications.

135 **2.0 Methodology**

136 *2.1 Materials*

137 Lignin was extracted from the oil palm empty fruit bunches and converted as
138 lignosulfonate using a method reported elsewhere [36]. Sodium polyacrylate
139 (PAAS) and Potassium Sulphate (K_2SO_4) were purchased from Sigma-Aldrich.
140 Polyimide film with a thickness of 0.127 mm was purchased from Avantis
141 Laboratory Supply. Deionized (DI) water was used in diluting chemicals and
142 preparing the samples.

143 *2.2 Preparation of OPL-LSG electrodes via laser scribing*

144 Lignin solution (20% v/v) was prepared by adding 20 g of lignin powder into 100 ml
145 of deionized water and stirring for 1 hr at room temperature. **The blade coating**
146 **technique was used to coat the lignin solution on a clean commercial polyimide**
147 **(PI) film at a blade-to-substrate distance of 500 μm .** The lignin-coated polyimide
148 substrate was utilized for laser scribing after being dried in an oven at 50°C for an
149 hour. The CO_2 laser engraving system (V-460, Universal Laser System,
150 Scottsdale, Arizona, USA), with a maximum power of 30 W and a **speed of 1000**
151 **mm/s**, was used to print the interdigitated electrode. In this work, the applied laser
152 power and the moving laser speed were varied from 50 % to 70 % and 30 % to
153 50 %, respectively. **Power and speed beyond this range was not considered since**

154 they are known to either fail to produce graphene or severely damage the flexible
155 polymer substrate. The untreated lignin layer was then dissolved using a water
156 bath, and the remaining water was dried overnight in an oven. OPL-LSG X Y
157 represents oil palm lignin laser-scribed graphene synthesized at a laser power of
158 X% and a speed of Y%. For example, LSG derived at a laser power of 50% and
159 laser speed of 50% is denoted as OPL-LSG 5050. Additionally, this work used an
160 ordinary polyimide derived laser scribed graphene (P-LSG X Y) as a control.

161 2.3 Assembly of OPL-LSG-based microsupercapacitors

162 The CorelDRAW software was used to create a CAD drawing of the desired
163 interdigitated electrode before the laser irradiation. The microsupercapacitor's
164 current collectors were coupled with conducting wire and covered with silver paste.
165 PAAS/K₂SO₄ (0.5 M) gel electrolyte was coated on top of 6 pairs of microelectrodes
166 with a total area of 0.5 cm² to complete the assembly of microsupercapacitor. Here,
167 PAAS/ K₂SO₄ (0.5 M) was made by mixing 10 ml of 0.5 M K₂SO₄ solution with 1g
168 of sodium polyacrylate salt until a gel-like material was created. Before obtaining
169 the electrical reading, the developed microsupercapacitor was wrapped in
170 polyimide tape and allowed to immerse the microelectrodes in the gel electrolyte
171 for a few hours, to ensure complete diffusion of electrolyte into electrode material.

172 2.4 Materials characterization

173 A square sample measuring 1 cm² in the area has been laser scribed for each of
174 the samples to measure the surface morphology of the material by Field Emission
175 Scanning Electron Microscopy, FESEM (TESCAN CLARA, UHR SEM), and
176 Scanning Electron Microscopy, SEM (Zeiss Evo LS15) analysis. A few laser-
177 scribed samples were scratched and mixed in ethanol to study the material's
178 microstructure through Transmission Electron Microscopy, TEM (Hitachi, HT7830).
179 The sheet resistance of 1 cm² OPL-LSG square samples was measured using van
180 der Pauw (VDP) technique using Ecopia, HMS-5000. The Raman spectra of 1 cm²
181 OPL-LSG square samples were obtained from Raman Spectrometer (DXR3
182 microscope) at 532 nm wavelength. The detailed calculation to identify crystalline
183 size (L_a) for Raman readings is included in the supplementary material. The
184 specific surface area (SSA) was computed using the Brunauer-Emmett-Teller
185 (BET) technique, and pore size distributions were assessed using the Barrett

186 Joyner Halenda (BJH) model on the **N₂ adsorption-desorption isotherms (Tristar**
187 **3020 Plus) for the black powder scratched from laser scribed material.** The surface
188 **material composition of the black powder** was examined using X-ray photoelectron
189 spectroscopy (XPS) by Thermo Scientific K-Alpha. The functional groups before
190 and after laser scribing were examined using Fourier-transform infrared
191 spectroscopy (FTIR) by Perkin Elmer, Spectrum One. The X-ray diffraction (XRD)
192 experiments **were carried out using an X-ray diffractometer for the black powder**
193 by PANalytical X'Pert3 Powder, with an exposure duration of two seconds per step
194 and a step size of 0.02 degrees per step.

195 *2.5 Electrochemical characterization*

196 Cyclic voltammetry (CV), galvanostatic charge-discharge (GCD), and
197 electrochemical impedance spectroscopy (EIS) of fabricated microsupercapacitor
198 were measured in a two-electrode system using an electrochemical workstation by
199 (Metrohm, Autolab). **A voltage window of -0.4 V to 0.4 V with a scan rate ranging**
200 **from 10-200 mVs⁻¹ was used in Cyclic voltammetry (CV) measurement.** Current
201 densities ranging from 0.08 – 1.0 mAcm⁻² were used in galvanostatic
202 charge/discharge (GCD) measures. EIS measurements with an amplitude of 10
203 mV were recorded in the frequency range of 0.01 Hz to 1E+5 Hz. The detailed
204 calculations of electrochemical readings are given in the supplementary material.

205

206 **3.0 Results and Discussion**

207 *3.1 Fabrication and characterization of OPL-LSG electrodes*

208 Fig. 1 depicts the schematic for constructing an OPL-LSG flexible
209 microsupercapacitor, primarily using laser scribing. **The lignin solution was first drop-**
210 **cast onto the polyimide sheet.** Following an hour of drying, the sample was laser
211 scribed using a CO₂ laser to generate the desired microelectrode pattern. Before
212 integrating the sample into a pliable micro supercapacitor, untreated lignin molecules
213 were removed using a water bath. The inset on the left indicates that graphene derived
214 from lignin has been functionalized by the assembled electrode. In contrast, the inset
215 on the right displays the FESEM images of highly porous 3D laser-scribed graphene
216 (LSG).

217 The surface morphology of OPL-LSG electrodes fabricated at various laser speeds
218 and power was observed using a Field Emission Scanning Electron microscope
219 (FESEM). **Notably, the highly porous structure of graphene is similar to previously**
220 **developed lignin** [22], [32], [37]. Low-resolution images of the laser-irradiated
221 graphene from OPL-LSG 7030 and OPL-LSG 5050 show a 3D wave-like porous
222 structure (Fig. 2a, d). In contrast, the magnified images reveal a beehive-like structure
223 with numerous macropores and mesopores induced by the discharge of gaseous
224 products such as H₂O and CO₂ (Fig. 2b, e) [21], [32]. The pores enhance the material's
225 surface area, which may promote electrolyte diffusion during the electrochemical
226 process [21], [22]. However, it is important to note that the electrode material produced
227 with a higher laser speed and lower laser intensity **has gaps exposing the polyimide**
228 **layer** between them that impede the flow of charge during the electrochemical process,
229 as shown in both views (Fig. 2d-e). Upon examining the cross-sectional view of the
230 electrode material, it is evident that OPL-LSG 7030 has a **higher average thickness** of
231 41.7 μm, densely packed with a foam-like material for better electrical continuity (Fig.
232 2c, f). Identical results were observed in polyimide graphene by varying the laser
233 power and speed (Fig. S1, S2). **Further evidence for the transformation of lignin into**
234 **porous graphene appears in the SEM images of lignin acquired before and after laser**
235 **scribing on top of plain ITO glasses (Fig. S3, S4).**

236 Fig. 2g shows Transmission Electron Microscopy (TEM) images of single-layer and
237 wrinkled multilayer graphene flakes. **OPL-LSG 7030 in Fig. 2g-h, show a variety of**
238 **pores ranging from meso to macro that would provide a larger surface area for**
239 **electrolyte ion facilitation** [32]. With a d-spacing (interlayer spacing) that correlates to
240 (002) lattice plane of 0.367nm, OPL-LSG 7030 has enhanced the incorporation and
241 removal of electrolyte ions within the graphene electrode (Fig. 2i) [7]. Additionally, the
242 existence of polycrystalline electrode material with a prominent ring reflects the (002)
243 plane, as shown by the selected area electron diffraction (SAED) pattern in Fig. 2i.
244 OPL-LSG 7030 has the lowest average sheet resistance, measuring 368
245 ohms/square, in line with FESEM findings and Van der Pauw measurements (Fig. 3a).
246 The findings demonstrate that resistance decreases as laser power increase and
247 speed decrease. Polyimide sheet coated with lignin performed as an insulator with a
248 notable resistance level.

249 Raman spectroscopy was used to examine the impact of varying laser power (fixed
250 speed at 30%) and speed (fixed power at 70%) on the synthesis of graphene. Based
251 on the results (Fig. 3b-c), all the samples exhibit three distinctive peaks located at
252 $\sim 1350\text{ cm}^{-1}$, $\sim 1585\text{ cm}^{-1}$, and 2690 cm^{-1} , which are associated with the D (triggered by
253 lattice defects and vacancies of sp^2 carbon bonds), G (sp^2 hybridized carbon), and 2D
254 (stacked layer of graphene) bands, respectively [17], [33]. The adequate thickness
255 and the structural disorder of the graphene were observed using the ratios of the I_G/I_{2D}
256 and I_D/I_G [32], [33]. Although a slight gain can be noticed initially, the I_D/I_G and I_G/I_{2D}
257 ratios of various laser powers and speeds progressively declined and reached
258 saturation (Fig. 3d-e). OPL-LSG 7030 is higher quality graphene than previous
259 variations, with the fewest graphene layers. At the same time, further increases in laser
260 power (80%) and decreases in laser speed (20%) deteriorate the polymer substrate
261 [22]. OPL-LSG 7030 has an average crystalline size of $\sim 31\text{ nm}$, 10 nm greater than the
262 P-LSG 7030 (Fig. S5). In addition, Raman spectra of graphene derived from lignin-
263 coated ITO glasses without polyimide indicate the presence of graphene by showing
264 similar D, G, and 2D peaks with I_G/I_{2D} and I_D/I_G values of 0.998 and 1.007 (Fig. S6).
265 As a result, increasing laser power while lowering laser speed is required to achieve
266 high-quality graphene from OPL.

267 The FTIR spectra in Fig. 3f represent the bonding states of graphene synthesized from
268 polyimide and oil palm lignin, respectively. The OH stretching is the broadest and most
269 dominating peak at $\sim 3400\text{ cm}^{-1}$. Compared to polyimide graphene, lignin graphene
270 shows a sharper C-O peak at 1070 cm^{-1} . Moreover, a small peak could be observed
271 at $\sim 1045\text{ cm}^{-1}$, representing the hydroquinone functionality group derived from lignin
272 [34], [38]. Besides that, carbon-carbon double bond and aromatics could be observed
273 for graphene at 1628 cm^{-1} and 1524 cm^{-1} [23]. The remaining absorption peaks, which
274 are situated at 2945 cm^{-1} , 2352 cm^{-1} , 1731 cm^{-1} , and 1406 cm^{-1} , are associated with
275 C-H stretching, O=C=O (CO_2), C=O bonds, and C-O stretching [23], [39]–[41]. Fig. S7
276 shows that the three peaks of pristine lignin, linked to S=O, C-O, and hydroquinone
277 and situated at 1195, 1112, and 1045 cm^{-1} , were significantly reduced with the
278 conversion of lignin into graphene [37], [40]–[42].

279 The N_2 adsorption and desorption isotherms were performed on OPL-LSG 7030 to
280 investigate the porosity of the sample (Fig. 4a). The type-III hysteresis loop that is
281 shown on the plotted isotherm confirms the presence of the meso- and macropores in

282 graphene, which would facilitate ion diffusion at high current densities [43], [44].
283 Notably, type III does not occur frequently, primarily because of the water vapor on the
284 basal plane of fabricated graphene [45]. Accordingly, OPL-LSG 7030 has a Brunauer-
285 Emmett-Teller (BET) specific surface area (SSA) of $37.13 \text{ m}^2\text{g}^{-1}$ and a pore volume of
286 $0.125 \text{ cm}^3\text{g}^{-1}$. The average pore diameter for OPL-LSG 7030, which consists of a
287 combination of meso and macro pores, is 11.69 nm, as further evidenced in Fig. 4b.
288 A variety of pores, from micro to macro, enhance the overall electrochemical
289 performance of OPL-LSG 7030 by expanding its surface area, accelerating electrolyte
290 ion movement, and retaining ions in a buffer zone [7].

291 To further examine the elemental compositions and chemical structures, XPS
292 measurements were performed for OP lignin and OPL-LSG 7030. The XPS survey
293 spectra of the samples are shown in Fig. 4c, which revealed three primary peaks at
294 285.3 eV, 531.3 eV, and 1072 eV, corresponding with C 1s, O 1s, and Na 1s,
295 respectively [22]. The Na peaks are primarily caused by residues from using NaOH
296 during the lignin extraction process [17], [22]. It is apparent that after laser scribing, C
297 1s intensity increased while Na 1s and O 1s intensity decreased, indicating that laser
298 scribing transforms lignin into high-quality graphene. The high-resolution C 1s XPS
299 peak were further divided into four components, which are located at $\sim 284.5 \text{ eV}$,
300 $\sim 285.2 \text{ eV}$, $\sim 286.2 \text{ eV}$, and $\sim 290.0 \text{ eV}$, respectively, for the C-C/C=C bond, the C-O
301 bond, the C=O bond, and the O-C=O bond [22], [46]. (Fig. 4d, S8). Furthermore, the
302 O 1s XPS spectra of both samples were broken down into three subpeaks, namely
303 C=O ($\sim 532.4 \text{ eV}$), C-O ($\sim 533.8 \text{ eV}$), and -COOH ($\sim 536.2 \text{ eV}$) [7], [47] (Fig. 4e, S9).
304 The greater laser power utilized to carbonize the lignin coated on polyimide increased
305 the local temperature and removed the impurities such as Na that may
306 impact electrochemical performance [32], [48]. Fig. 4e and S9 further demonstrate
307 how the dominance of the C-O bond has weakened and been superseded by the other
308 bonds made available throughout the graphitization process.

309 XRD analysis was used to examine the structural analysis of fabricated graphene (Fig.
310 4f). Both samples exhibited one broader peak at 25.9° and a minor peak at 43° , as
311 shown in the previous SAED image. The (002) peak's asymmetry is connected to the
312 OPL-LSG's minor disordered amorphous structure, while the sharp peak confirms the
313 fact that lignin was converted into a highly crystallized graphitic structure [23], [32]. In
314 comparison to OPL-LSG 5050, OPL-LSG 7030 has significantly more evident and

315 unambiguous (002) and (100) peaks in the analysis [23]. Due to the presence of
316 Na_2SO_4 (a salt produced by the soda pulping process), OPL-L5050 from lignin
317 precursor showed additional peaks at $\sim 34^\circ$, which is further supported by the
318 presence of Na impurities found in XPS survey spectra [49], [50]. Inadequate
319 graphitization at low laser power has prevented the salt from protruding and being
320 washed away [32].

321 3.2 Electrochemical performance OPL-LSG-based films

322 All the prepared samples were assembled into a complete electrode with the addition
323 of PAAS/ K_2SO_4 electrolyte and conducting wires, and the electrochemical
324 characteristics of each electrode were assessed. A potential window ranging from -
325 0.4V to 0.4V has been utilized for cyclic voltammetry (CV) and Galvanostatic Charge-
326 Discharge (GCD) curves. Fig. 5a shows the CV curves of a microsupercapacitor with
327 OPL-LSG at various laser power and speed. A pair of distinct redox peaks can be
328 observed from the CV curves due to quinone/hydroquinone groups (aromatics
329 compound) from oil palm lignin in the fabricated laser-scribed graphene observed in
330 FTIR. Hydroquinone offers pseudocapacitance for graphene and enriches its overall
331 functionality by storing and releasing electrons during the electrochemical process,
332 [51], [52]. The estimated areal capacitance (measured in mFcm^{-2}) at 50 mVs^{-1} of OPL-
333 LSG 5030, OPL-LSG 6030, OPL-LSG 7030, OPL-LSG 7040, and OPL-LSG 7050 are
334 1.57, 2.75, 3.15, 2.84 and 1.72, respectively. OPL-LSG 7030 exhibits the highest areal
335 capacitance at 50 mVs^{-1} among the different laser powers and speeds considered,
336 indicating that the electrochemical performance has been improved by the densely
337 packed graphene layer from FESEM images.

338 To further examine the samples, a detailed CV test was performed on the produced
339 samples at various scan rates, ranging from 10 mVs^{-1} to 200 mVs^{-1} (see Fig. 5b, S10).
340 It was apparent that at increased scan speeds, cathodic peak transitions to the positive
341 side and anodic peak transitions to the negative side of the plot were all caused by the
342 internal resistance of the electrode [53]. Fig. 5c depicts the relationship between
343 samples of varying laser power and speed performed at various scan rates with the
344 areal capacitance determined from CV curves. OPL-LSG 7030 surpasses all
345 comparable samples at lower scan rates yet showed average values at higher scan
346 rates due to the limited access time of electrolyte ions entering the thicker graphene

347 layer. OPL-LSG 7030's excellent areal capacitance value of 8.88 mFcm^{-2} at a scan
348 rate of 10 mVs^{-1} indicates that higher laser power and lower laser speed resulted in
349 higher quality graphene.

350 Fig. 5d represents graphene's Galvanostatic Charge Discharge (GCD) curves
351 synthesized at various power and speed settings. The GCD curves' deviation from the
352 ideal triangular shape demonstrates a significant contribution from
353 pseudocapacitance, supporting earlier CV test results. OPL-LSG 7030 had the longest
354 discharge time, as indicated by the GCD profile measured at a current density of
355 0.1 mAcm^{-2} , which is consistent with the larger area from CV curves. When OPL-LSG
356 7030 was tested at current densities varying from 0.08 mAcm^{-2} to 1 mAcm^{-2} , similar
357 outcomes were seen, with excellent coulombic efficiency (Fig. 5e). It is clear that at
358 lower current densities, the electrolyte ions penetrate the interior plane of the electrode
359 material at a higher rate compared to higher current densities. The electrochemical
360 reading of the tested material is enhanced substantially at lower current density, with
361 reactivity from both inner and outer surfaces [53]. Fig. 5f compares the areal
362 capacitance obtained from the GCD profile of assembled graphene electrodes and
363 current density. The calculated areal capacitance (measured in mFcm^{-2}) of OPL-LSG
364 at a current density of 0.08 mAcm^{-2} with increasing laser power are 14.8, 25, and
365 30.77, while 30.77, 24.5, and 13 at increasing laser speeds. Once again, OPL-LSG
366 7030 proves its remarkable capacitance property at higher and lower current densities
367 at a **volumetric capacitance of 7.379 Fcm^{-3}** .

368 Fig. 6a and the inset illustrate the Nyquist plots of laser-scribed lignin at various power
369 and speed settings between 100 kHz and 10 mHz. Equivalent Series Resistance
370 (ESR) measures the surface resistance of the material in the high-frequency region,
371 where the plot intersects the X axis (real axis). In contrast, Warburg resistance is the
372 electrolyte ion diffusion into the pores of electrodes in a low-frequency region [6], [54].
373 Overall, all the displayed graphs exhibit the standard behaviour of porous carbon, with
374 steeper inclinations of more than 45° without a distinct semicircle appearance [55].
375 OPL-LSG 7030 recorded the lowest ESR with a value of $\sim 60.1 \Omega$, indicating capacitor
376 behaviour, while maintaining a close distance from the Y axis (imaginary axis). The
377 ESR values for other samples from OPL-LSG 5030 to OPL-LSG 7050 are 158Ω , 77.3
378 Ω , 91.1Ω , and 133Ω , respectively. The EIS data and other electrochemical

379 measurements provided additional evidence of the high conductivity of fabricated
380 graphene.

381 The electrochemical stability was assessed at a current density of 0.5 mAcm^{-2} to
382 examine the overall performance of synthesized graphene, as shown in Fig. 6b.
383 Notably, OPL-LSG 7030 has a cycling stability of 88.4% capacitance retention over
384 5000 cycles with an initial boost in capacitance retention due to an “activation process”
385 [56]. OPL-LSG 7030 retained 80.78% of its initial capacitance after 8000 cycles (Fig.
386 S11). The overall decline in the plot was primarily caused by graphene detachment on
387 the fabricated electrode, which reduced the active area of the electrode [57].
388 Additionally, OPL-LSG 7030 demonstrated the highest areal energy density of
389 $0.00176 \text{ mWhcm}^{-2}$ at volumetric energy density 0.422 mWhcm^{-3} , and the highest areal
390 power density of 0.25 mWcm^{-2} at volumetric power density 59.95 mWcm^{-3} due to the
391 thicker graphene layer's ability to facilitate greater ionic activity (Fig. 6c).

392 The fabricated OPL-LSG 7030 electrode was subsequently subjected to a bending
393 test to determine its pliability in a practical application. The CV test for 400 continuous
394 cycles at 50 mVs^{-1} demonstrated remarkable mechanical stability at the 400th bending
395 cycle (Fig. 6d), with capacitance retention of over 95% (Fig. 6e) in the observable
396 region. The Ragone plot in Fig. 6f depicts the relationship between the energy and
397 power density of fabricated graphene OPL-LSG 7030 which is comparable to most
398 previously reported laser-derived planar supercapacitors including 5B-LIG [58], M-
399 PBRV-RGO [59], LSG-P24 [32], and PA-PBI-LIG [60]. At the same time, OPL-LSG
400 7030 outperforms LIG [61] and 3D-GF [62] in terms of energy and power density.
401 Furthermore, Ph-ddm/LIG [63], KOH-LIG [64], LIG-B6 [43], LSG-P24-Au [32], sLIG-
402 O/S14 [33] and LWG-ONS 1.5 [7] microsupercapacitors possess an extraordinary
403 electrochemical result either due to the composite material formation, electrode
404 activation or heteroatom doping. Table S1 provides a comparison of the material
405 mentioned above.

406 **4.0 Conclusion**

407 In summary, lignin from oil palm EFB has been effectively transformed into a
408 functionalized graphene using a single-step and cost-effective laser scribing
409 technique. The resulting graphene possesses redox qualities as the laser scribing
410 process retained the properties of the aromatic compounds within the lignin. A high-

411 quality 3D graphene with a highly porous structure (meso and macro pores), reduced
412 resistance, larger surface area, and smaller pore size has been developed at a laser
413 power of 70 % and laser speed of 30 %. Most importantly, the resulting
414 microsupercapacitor exhibits improved electrochemical properties such as high areal
415 capacitance (30.77 mFcm^{-2}), decent energy/power densities ($0.00176 \text{ mWhcm}^{-2}$, 0.25
416 mWcm^{-2}), and comparable cyclic stability (88.4% for 5000 cycles) than those reported
417 in previous studies. The developed graphene electrode also displayed exceptional
418 mechanical durability at a scan rate of 50 mVs^{-1} following 400 bending cycles. **Doping**
419 **or incorporating nanocomposites could further enhance this green material's**
420 **performance for flexible microsupercapacitor applications.**

421

422 References

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620 Acknowledgements

621 The authors would like to thank Universiti Teknologi PETRONAS (UTP) and Yayasan
622 Universiti Teknologi PETRONAS (YUTP) for the financial support through YUTP-FRG
623 015CL0-454 for giving the opportunities to conduct the research in the Centre of
624 Innovative Nanostructures & Nanodevices (COINN). The appreciation also goes to all
625 the team members and staffs in the Department of Mechanical Engineering, UTP.

626 Figure legends

627 Fig. 1. Schematic illustration of the procedure to synthesize OPL-LSG electrodes.

628 Fig. 2. FESEM images of OPL-LSG 7030 at low (a) and high (b) magnification. Cross
629 sectional view of OPL-LSG 7030 (c). FESEM images of OPL-LSG 5050 at low (d) and
630 high (e) magnification. Cross sectional view of OPL-LSG 5050 (f). TEM images of
631 OPL-LSG 7030 at low (g) and high (h) magnification. TEM OPL-LSG 7030 with lattice
632 fringes (The inset represents SAED pattern) (I).

633 Fig. 3. Sheet resistance of OPL-LSG XY (a), Raman Spectra of OPL-LSG at different
634 laser power (b), and speed (c). Correlations of the ID/IG and IG/I2D obtained at
635 different laser power (d) and speed (e). FTIR of OPL-LSG 7030 and P-LSG 7030 (f).

636 Fig. 4. N₂ adsorption–desorption curves (a) and Pore size distribution (Inset:
637 amplifying pore size distribution) (b), for OPL-LSG 7030. XPS survey of lignin and
638 OPL-LSG 7030 (c), C 1s (d) and O 1s (e) of OPL-LSG 7030. XRD patterns of OPL-
639 LSG 7030 (f).

640 Figure 5 CV curves of; OPL-LSG XY at a scan rate of 50mVs⁻¹ (a) and OPL-LSG 7030
641 at various scan rates (b). Dependence of OPL-LSG XY capacitance at different scan
642 rates (c). GCD curves of; OPL-LSG XY at a current density of 0.1mAcm⁻² (d) and OPL-
643 LSG 7030 at various current densities (e). Correlation of OPL-LSG XY capacitance at
644 different current densities (f).

645 Figure 6 Nyquist plots of OPL-LSG XY (Inset: close up image of Nyquist plots and EIS
646 equivalent fitting circuit) (a). Cyclic stability of OPL-LSG 7030 (b). Correlation of energy
647 density and power density of OPL-LSG 7030 (c). CV curves of OPL-LSG7030 at
648 different bending cycles at 50mVs⁻¹ (d), Capacitance retention as a function of bending
649 cycle (Inset: Image of fabricated electrode during bending cycle) (e). Ragone plot for
650 comparison of energy density and power density of OPL-LSG 7030 with previously
651 reported supercapacitors (f).

652